

In-plane optical spectroscopy study on FeSe epitaxial thin film grown on SrTiO₃ substrate

R. H. Yuan, W. D. Kong, L. Yan, H. Ding, and N. L. Wang

Beijing National Laboratory for Condensed Matter Physics, Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China

(Received 10 March 2013; revised manuscript received 21 April 2013; published 29 April 2013)

We perform in-plane optical spectroscopy measurement on (001) FeSe thin film grown on SrTiO₃ substrate by the pulsed laser deposition method. The study indicates that the low-frequency conductivity consists of two Drude components: a broad one which takes up most of the spectral weight and a very narrow one roughly below 100–150 cm⁻¹. The narrow Drude component locates at so low frequencies that no such behavior was observed in iron pnictides. The overall plasma frequency is smaller than that of the FeAs-based compounds, suggesting a stronger correlation effect. Similar to iron pnictides, a temperature-induced spectral weight transfer is observed for FeSe. However, the relevant energy scale is smaller. Additionally, different from a recent angle-resolved photoemission spectroscopy measurement which revealed a spin density wave (SDW) order at low temperature for FeSe thin films grown on SrTiO₃ substrate, no signature of the band structure reconstruction arising from the formation of the SDW order is seen by optical measurement in the thick FeSe films.

DOI: [10.1103/PhysRevB.87.144517](https://doi.org/10.1103/PhysRevB.87.144517)

PACS number(s): 74.25.Gz, 74.70.Xa

The binary β -FeSe with the PbO structure is a key member of the family of high- T_c iron pnictide and chalcogenide superconductors. Its structure comprises stacks of edge-sharing FeSe₄ tetrahedra, similar to that of FeAs layers in the family of the iron pnictide superconductors, but lacking charge-reservoir layers. FeSe is thus the simplest Fe-based superconductor. The undoped FeSe exhibits superconductivity with $T_c = 8$ K.¹ Upon applying pressure, T_c dramatically rises to 37 K.² Recent studies on single layer FeSe thin film grown on SrTiO₃ substrate by the molecular beam epitaxy (MBE) method gave a hint that the T_c may even exceed 65 K.^{3–6} It is widely expected that investigation of binary FeSe will provide important clues to elucidate the superconducting mechanism of iron-based superconductors.

Theoretical calculations indicate that β -FeSe has a two-dimensional electronic structure similar to that of the Fe pnictides with cylindrical electron sections at the zone corner and hole sections at the zone center.^{7–9} Therefore, a similar spin density wave (SDW) order with a collinear antiferromagnetic structure was predicted.^{8,9} However, no experimental evidence for the presence of such SDW order was indicated in earlier experiments on polycrystalline samples.¹⁰ Single crystal growth of pure FeSe turns out to be difficult due to the very narrow phase formation range.^{11–15} The reported platelike single crystals grown by using ACl ($A = \text{Li, Na, and K}$) as flux usually contain secondary phase and quite often have a (101) surface,¹⁵ making it very difficult to study the intrinsic in-plane properties. As a result, pure FeSe was far less studied than other Fe-based superconductors, particularly by spectroscopic techniques. On the other hand, recent studies indicate that the epitaxial growth of thin films on SrTiO₃ substrate by MBE method could yield high-quality (001) FeSe samples,^{3–6} making them ideal for in-plane physical property characterizations. Interestingly, a very recent angle-resolved photoemission spectroscopy (ARPES) measurement⁶ on such FeSe thin films indicated that, except for the single layered FeSe film, the band structure near the Brillouin zone corner at low temperature is very similar to those observed for the parent compounds of FeAs-based systems, e.g., BaFe₂As₂ and NaFeAs in the SDW state,^{6,16} yielding evidence for

the formation of collinear antiferromagnetic structure. It was also found that the SDW order temperature decreases with increasing the thin film thickness, which was ascribed to the lattice relaxation effect. The thick thin films (over 35 monolayers) resemble the bulk samples but still have magnetic SDW ground state.⁶

Optical spectroscopy is a powerful bulk technique to investigate charge dynamics and band structure of materials as it probes both free carriers and interband excitations.¹⁷ In particular, it yields direct information about the formation of energy gaps. Infrared spectroscopy studies on the parent compounds of Fe pnictides (including systems in 122, 1111, and 111) provide clear evidence for the SDW gap formations in the ordered state.^{18–20} It would be very interesting to investigate the charge dynamic of binary β -FeSe to see if a similar gap opening effect exists by bulk probe technique. It should also be very helpful to compare the charge dynamical properties of FeSe with other Fe-pnictide and chalcogenide superconductors. In this work, we present an optical spectroscopy study on the in-plane properties of β -FeSe thin film grown on SrTiO₃ substrate.

The thin-film samples used in this study were obtained by the pulsed laser molecular beam epitaxy on a (001) SrTiO₃ substrate. The advantage of this thin-film growth technique is that it is easy to grow relatively thick films. The laser energy, repetition, substrate temperature, and pressure are 250 mJ/pulse, 10 Hz, 400 °C and 3×10^{-8} torr, respectively. The films had been deposited for 7 h, resulting in a thickness of 950 nm. The epitaxy growth of the film is confirmed by the x-ray diffraction measurement, as shown in Fig. 1. Besides the diffraction lines from the SrTiO₃ substrate, only FeSe(001) diffraction peaks are present in the pattern. The pattern indicates that the film has an ab -plane surface. The composition ratio of Fe:Se was confirmed to be 1:1 by the scanning electron microscope equipped with energy dispersive x-ray analysis. Note that the epitaxial growth of the film depends strongly on the substrate. If the film is grown on a MgO substrate, the film would have a (101) crystal surface.²¹

Figure 2 shows the in-plane resistivity measured in a Quantum Design PPMS system. The resistivity shows a

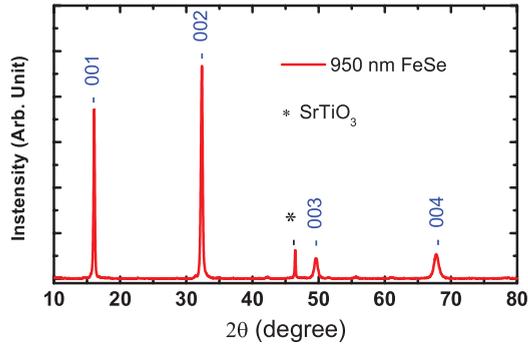


FIG. 1. (Color online) X-ray diffraction pattern of the 950-nm FeSe thin film grown on (001) SrTiO₃ substrate.

superconducting transition temperature of about 7 K, close to the bulk T_c . The inset shows the temperature-dependent derivative of the resistivity. A kink feature could be observed near 100 K, being similar to the report on polycrystalline samples,²² which were usually linked with the weak structural distortion. A further increase of the slope is seen near 50 K, as evidenced more clearly from the derivative plot.

The optical reflectance measurements were performed on a combination of Bruker IFS 80v/s and 113v spectrometers in the frequency range from 20 to 25 000 cm⁻¹. An *in situ* gold and aluminum overcoating technique was used to get the reflectivity $R(\omega)$. The real part of conductivity $\sigma_1(\omega)$ is obtained by the Kramers-Kronig transformation of $R(\omega)$.

Figure 3 shows the room temperature optical reflectance $R(\omega)$ (upper panel) and conductivity $\sigma_1(\omega)$ (lower panel). No leakage from the SrTiO₃ substrate is detected. The overall spectral line shapes are similar to iron pnictides, e.g., BaFe₂As₂. As a comparison, we have included the room temperature reflectance data of BaFe₂As₂ in the figure.¹⁸ The reflectance drops quickly with frequency at the low- ω region, then merges into the high values of a background contributed mostly from the incoherent carriers and interband transitions. However, a clear difference between FeSe and BaFe₂As₂ could be identified: The reflectance of FeSe at the low frequency

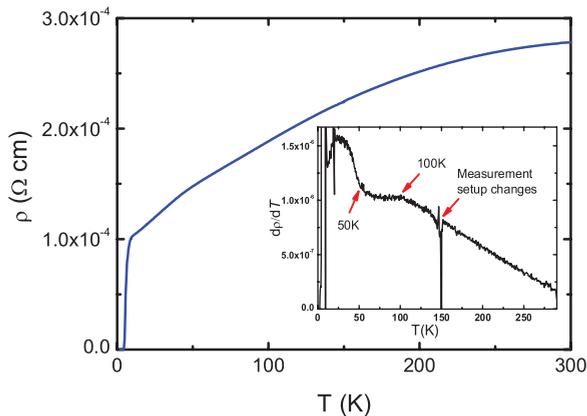


FIG. 2. (Color online) Temperature-dependent resistivity of FeSe film. The inset is the derivative of the resistivity as a function of temperature.

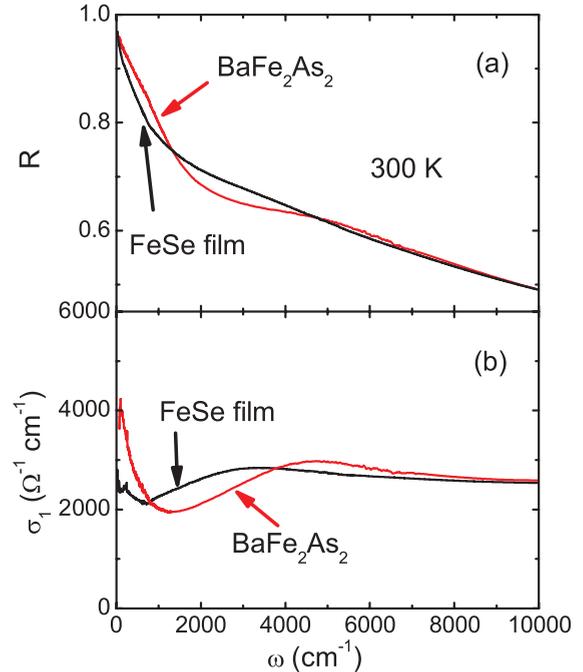


FIG. 3. (Color online) The room temperature optical reflectance $R(\omega)$ (a) and conductivity $\sigma_1(\omega)$ (b) over broad frequencies. Data from the BaFe₂As₂ (Ref. 18) are also shown for comparison.

range (below 1200 cm⁻¹) is apparently lower than that of BaFe₂As₂, but reversed at higher frequencies. This leads to smaller free carrier spectral weight but enhanced spectral weight in the frequency range between 1000 and 4000 cm⁻¹ in $\sigma_1(\omega)$.

The temperature-dependent optical reflectance $R(\omega)$ and conductivity $\sigma_1(\omega)$ are shown in Fig. 4. The upper panels [(a) and (b)] show the spectra over broad frequencies (up to 8000 cm⁻¹, ~ 1 eV); the lower panels [(c) and (d)] are the expanded plots at low frequencies. From the lower panels, we can see a phonon mode at 240 cm⁻¹ at room temperature, which shifts slightly to higher frequency at low temperature (248 cm⁻¹ at 8 K) and also becomes more visible. The phonon mode is commonly seen in the in-plane infrared measurement on FeAs(Se)-based single crystal samples and ascribed to the in-plane displacements of Fe-As(Se) atoms.²³ The mode appears at a higher frequency than that observed at 187 cm⁻¹ for Fe_{1.03}Te and 204 cm⁻¹ for FeTe_{0.55}Se_{0.45} samples.²⁴ This is due to the reduced mass of the Se atom as compared with the Te atom. For FeAs-based compounds, the mode appears at a slightly higher energy scale, e.g., 253 cm⁻¹ for BaFe₂As₂ at 300 K.^{18,23} The clear observation of this in-plane phonon mode is an indication of the good quality of the sample.

It is tempting to estimate the Drude weight or plasma frequency of the FeSe sample and compare it with other iron pnictides/chalcogenides. The $\sigma_1(\omega)$ spectrum shows very weak frequency-dependent behavior below roughly 700–800 cm⁻¹. However, a sharp and narrow component develops below 100–150 cm⁻¹ [see Fig. 4(d)]. The spectra are more similar to FeTe or FeTe_{0.55}Se_{0.45} single crystals^{24,25} at low temperature than BaFe₂As₂ or other Fe pnictides.^{18–20,23} If we consider only the sharp and narrow component below 100–150 cm⁻¹ as the

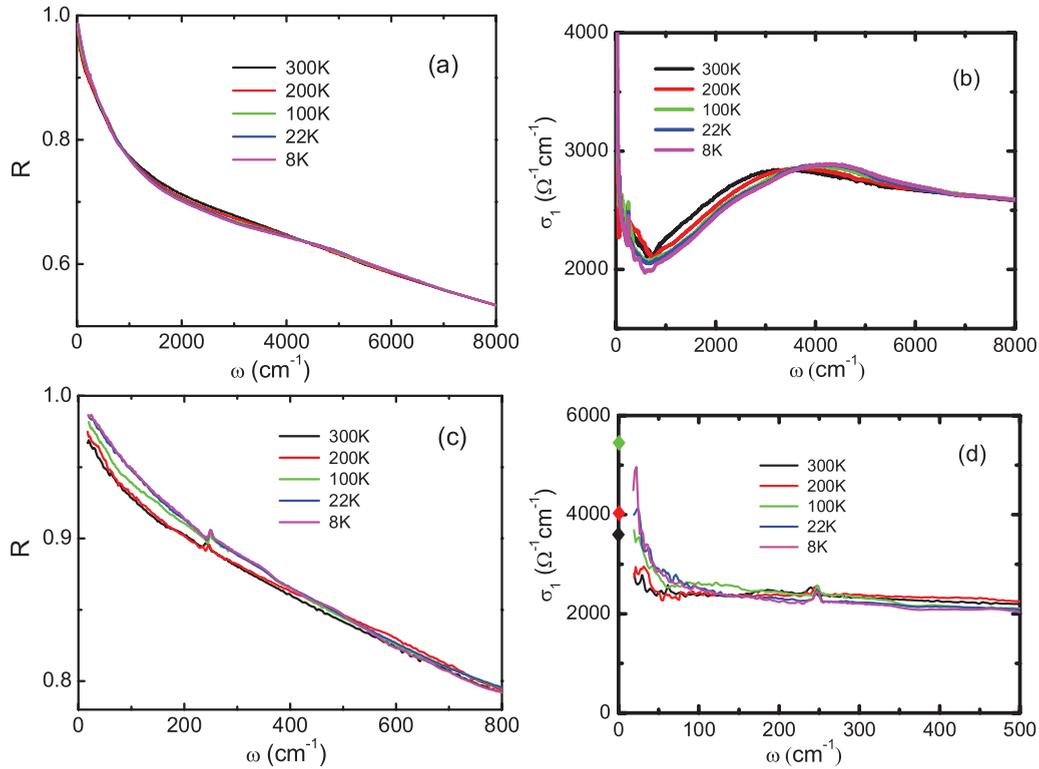


FIG. 4. (Color online) The reflectance $R(\omega)$ and conductivity spectra $\sigma_1(\omega)$ of FeSe film at different temperatures. Upper panels: the spectra over broad frequencies up to 8000 cm^{-1} . Lower panels: the expanded plot of $R(\omega)$ and $\sigma_1(\omega)$ in the low-frequency region. The dc conductivity values at three temperatures—300, 200, and 100 K—were added.

Drude component, and the slowly decreasing $\sigma_1(\omega)$ spectrum above it as incoherent, we may underestimate the plasma frequency. So we assume two components contribute to the Drude spectral weight: an extremely narrow one below 150 cm^{-1} and a broad one contribute to the rather slowly decreasing conductivity between 150 and 800 cm^{-1} . Then, we can analyze the $\sigma_1(\omega)$ data using a Drude-Lorentz model in a way similar to what we did for 122-type crystals,¹⁸

$$\epsilon(\omega) = \epsilon_\infty - \sum_i \frac{\omega_{p,i}^2}{\omega_i^2 + i\omega/\tau_i} + \sum_j \frac{\Omega_j^2}{\omega_j^2 - \omega^2 - i\omega/\tau_j}, \quad (1)$$

where ϵ_∞ is the dielectric constant at high energy, and the middle and last terms are the Drude and Lorentz components, respectively.

We find that the optical conductivity data could be reasonably reproduced by this approach. Figure 5 shows the $\sigma_1(\omega)$ data at 300 K together with the components in the Drude-Lorentz analysis. Panel (a) shows the spectral data at low frequencies, while panel (b) shows the data over broad frequencies up to 8000 cm^{-1} . The parameters of two Drude components are $\omega_{p1} = 1300 \text{ cm}^{-1}$, $1/\tau_1 = 20 \text{ cm}^{-1}$ and $\omega_{p2} = 10900 \text{ cm}^{-1}$, $1/\tau_1 = 840 \text{ cm}^{-1}$ at 300 K. At 8 K, those parameters become $\omega_{p1} = 2700 \text{ cm}^{-1}$, $1/\tau_1 = 16 \text{ cm}^{-1}$ and $\omega_{p2} = 10600 \text{ cm}^{-1}$, $1/\tau_1 = 830 \text{ cm}^{-1}$. Then, the overall plasma frequency ω_p could be calculated as $\omega_p = (\omega_{p1}^2 + \omega_{p2}^2)^{1/2} \approx 11000 \text{ cm}^{-1}$ (1.36 eV) for different temperatures. This value is somewhat smaller than that in

BaFe₂As₂ (about 1.6 eV).¹⁸ In density functional theory (DFT) calculations, the plasma frequency of FeSe is slightly higher than 3 eV.²⁶ In comparison with this value, we find the band renormalization factor $\omega_{p,\text{DFT}}^2/\omega_{p,\text{exp}}^2 \sim 5$, which is larger than BaFe₂As₂. On the other hand, the value is close to that obtained by a combination of density functional theory and dynamical mean field theory where electron correlation effect has been properly taken into account.^{26,27} The agreement between experimentally obtained values and those from such theoretical calculations for Fe pnictides was already well documented.²⁶

The plasma frequency could be alternatively estimated by summarizing the low- ω spectral weight, $\omega_p^2 = 8 \int_0^{\omega_c} \sigma_1(\omega) d\omega$. The integration up to ω_c should cover all of the spectrum contributed by the free carriers but still below the interband transition. Usually, one takes ω_c at the frequency where the $\sigma_1(\omega)$ shows a minimum, then we expect that there is a balance between the Drude component tail and the onset part of the interband transition. Taking $\omega_c \approx 800 \text{ cm}^{-1}$, we get $\omega_p \approx 8400 \text{ cm}^{-1}$. This value is smaller than the value obtained from the above two Drude component analysis. This is because the tail of the broad Drude component is not balanced by the onset part of the Lorentz component, as can be seen in Fig. 5(b). Therefore, taking ω_c at the energy that $\sigma_1(\omega)$ shows a minimum in the present case still leads to an underestimation of plasma frequency.

Although the presence of multiple Drude components appears to be the generic property of iron pnictides/chalcogenides, the development of a very narrow

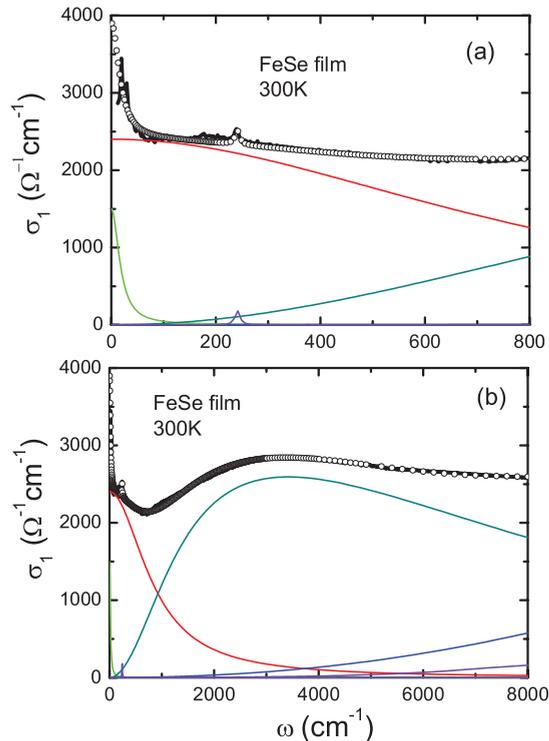


FIG. 5. (Color online) The experimental data of $\sigma_1(\omega)$ at 300 K FeSe film and the Drude-Lorentz fit results. (a) The data and fitting curve at low-frequency range, and (b) the data and fitting curves at broad frequency range.

and sharp Drude component at such low frequencies was not seen in iron pnictide systems.^{18–20,23} It suggests that in certain peculiar bands of FeSe the quasiparticles experience extremely small scattering at low temperature.

Another prominent feature in optical conductivity [Fig. 4(b)] is that the low-frequency spectral weight is transferred to the high-energy region above 4000 cm^{-1} with decreasing temperature. Such spectral weight transfer feature was also observed in iron-pnictide systems,^{28,29} and was referred to as a “high-energy pseudogap” structure.²⁸ But the relevant energy scale is smaller for FeSe. For BaFe_2As_2 , the spectral weight was transferred to a region above 5000 cm^{-1} . The temperature-induced spectral weight transfer was ascribed to the electron correlation effect, in particular, to the Hund’s coupling effect between itinerant Fe 3d electrons and localized Fe 3d electrons in different orbitals.^{28,29} So, the spectral weight transfer represents the redistribution of the spectral weight between different 3d bands. It is interesting to note that FeSe as well as other iron-calcogenide compounds have a stronger electron correlation effect, leading to a larger local moment of $2\mu_B$ (Ref. 30) and higher band renormalization factor. However, the energy scale of temperature-induced spectral weight transfer in optical conductivity appears to be smaller. This puzzling phenomenon reflects the complex interplay effect from correlations/Hund’s couplings and kinetic energy of electrons in FeSe(As) systems and needs to be further explored.

As we mentioned above, a recent ARPES measurement on (001) FeSe thin film⁶ indicated signature of SDW order, in a way very similar to the parent compound of Fe pnictides,

e.g., BaFe_2As_2 . It would be highly desirable to examine this possibility by optical measurement. From Fig. 4(c), we find that the reflectance $R(\omega)$ at low frequency displays a monotonous increase with decreasing temperature without showing any specific suppression arising from energy gap formation. At two lowest measurement temperatures, i.e., 8 and 22 K, the reflectance seems to develop a stronger upward curvature below the phonon mode frequency, being consistent with the even faster decrease of the dc resistivity below 50 K. Nevertheless, no energy gap signature could be identified from the reflectance spectra. As a result, in the $\sigma_1(\omega)$ spectra, we observe a further narrowing of Drude component at low temperature. Since the infrared measurement is a bulk detection technique and infrared experiments on all parent compounds of iron pnictides (including 122, 111, and 1111 systems) clearly revealed the formation of energy gaps in the SDW state or even below the structural distortion, our measurement indicates that the superconducting bulk FeSe film is not in the SDW state.

It is worth noting that an earlier pump-probe experiment on (101) FeSe film indicated a gapped relaxation of excited quasiparticles below roughly 100 K.²¹ However, the extraction of an energy gap from this technique is indirect. The quasiparticles were excited by the pump laser (3.1 eV) to states far above the Fermi level, then the transient reflection was probed at the energy of 1.55 eV on those excited quasiparticles in the relaxation process towards their equilibrium state. The probe is different from the present optical spectroscopy measurement in the equilibrium state. It is also worth noting that the ARPES experiment on FeSe thin films did not reveal any sign of band folding upon the development of SDW order.⁶ Although the band structure near the zone corner at low temperature is similar to BaFe_2As_2 and NaFeAs in the SDW state, the band structure near the zone center does not change with temperature. As suggested, this could be caused by the lack of long-range coherence.⁶ A possible scenario to reconcile the different measurements is that there is no static SDW order in FeSe films in its equilibrium state, but fluctuated SDW order at a very short time scale may exist, which has been probed by the ARPES and pump-probe measurements.

In summary, we have investigated the *ab*-plane optical properties of FeSe films grown on SrTiO_3 substrate and compared with Fe pnictides, e.g., BaFe_2As_2 . The low-frequency conductivity spectrum consists of two components: a broad one which takes up most of the spectral weight and a narrow one roughly below 100–150 cm^{-1} . The narrow Drude component locates at so low frequencies that no such behavior was observed in iron-pnictide systems. The study revealed a smaller plasma frequency and enhanced renormalization effect in FeSe. Furthermore, different from the recent ARPES measurement which revealed a SDW order at low temperature for relatively thick thin films grown on SrTiO_3 substrate, the present optical measurement does not yield any sign of the energy gap formation in the thick film.

This work was supported by the National Science Foundation of China and the 973 project of the Ministry of Science and Technology of China (Grants No. 2011CB921701 and No. 2012CB821403).

- ¹Fong-Chi Hsu *et al.*, *Proc. Natl. Acad. Sci. USA* **105**, 14262 (2008).
- ²S. Medvedev *et al.*, *Nat. Mater.* **8**, 630 (2009).
- ³Q. Y. Wang *et al.*, *Chin. Phys. Lett.* **29**, 037402 (2012).
- ⁴D. Liu *et al.*, *Nat. Commun.* **3**, 931 (2012).
- ⁵S. He *et al.*, arXiv:1207.6823.
- ⁶S. Y. Tan *et al.*, arXiv:1301.2748.
- ⁷A. Subedi, L. Zhang, D. J. Singh, and M. H. Du, *Phys. Rev. B* **78**, 134514 (2008).
- ⁸F. Ma, W. Ji, J. Hu, Z.-Y. Lu, and T. Xiang, *Phys. Rev. Lett.* **102**, 177003 (2009).
- ⁹C. Y. Moon and H. J. Choi, *Phys. Rev. Lett.* **104**, 057003 (2010).
- ¹⁰For a review, please see D. J. Singh, *Sci. Technol. Adv. Mater.* **13**, 054304 (2012).
- ¹¹S. B. Zhang *et al.*, *Supercond. Sci. Technol.* **22**, 015020 (2009).
- ¹²T. M. McQueen, Q. Huang, V. Ksenofontov, C. Felser, Q. Xu, H. Zandbergen, Y. S. Hor, J. Allred, A. J. Williams, D. Qu, J. Checkelsky, N. P. Ong, and R. J. Cava, *Phys. Rev. B* **79**, 014522 (2009).
- ¹³M. K. Wu *et al.*, *Physica C* **469**, 340 (2009).
- ¹⁴U. Patel *et al.*, *Appl. Phys. Lett.* **94**, 082508 (2009).
- ¹⁵R. Hu, H. Lei, M. Abeykoon, E. S. Bozin, S. J. L. Billinge, J. B. Warren, T. Siegrist, and C. Petrovic, *Phys. Rev. B* **83**, 224502 (2011).
- ¹⁶L. X. Yang, Y. Zhang, H. W. Ou, J. F. Zhao, D. W. Shen, B. Zhou, J. Wei, F. Chen, M. Xu, C. He, Y. Chen, Z. D. Wang, X. F. Wang, T. Wu, G. Wu, X. H. Chen, M. Arita, K. Shimada, M. Taniguchi, Z. Y. Lu, T. Xiang, and D. L. Feng, *Phys. Rev. Lett.* **102**, 107002 (2009).
- ¹⁷M. Dressel and G. Gruner, *Electrodynamics of Solids* (Cambridge University Press, Cambridge, UK, 2002).
- ¹⁸W. Z. Hu, J. Dong, G. Li, Z. Li, P. Zheng, G. F. Chen, J. L. Luo, and N. L. Wang, *Phys. Rev. Lett.* **101**, 257005 (2008).
- ¹⁹T. Dong, Z. G. Chen, R. H. Yuan, B. F. Hu, B. Cheng, and N. L. Wang, *Phys. Rev. B* **82**, 054522 (2010).
- ²⁰W. Z. Hu, G. Li, P. Zheng, G. F. Chen, J. L. Luo, and N. L. Wang, *Phys. Rev. B* **80**, 100507 (2009).
- ²¹Y.-C. Wen, K.-J. Wang, H.-H. Chang, J.-Y. Luo, C.-C. Shen, H.-L. Liu, C.-K. Sun, M.-J. Wang, and M.-K. Wu, *Phys. Rev. Lett.* **108**, 267002 (2012).
- ²²Y. J. Song *et al.*, *J. Korean Phys. Soc.* **59**, 312 (2011).
- ²³A. Akrap, J. J. Tu, L. J. Li, G. H. Cao, Z. A. Xu, and C. C. Homes, *Phys. Rev. B* **80**, 180502(R) (2009).
- ²⁴C. C. Homes, A. Akrap, J. S. Wen, Z. J. Xu, Z. W. Lin, Q. Li, and G. D. Gu, *Phys. Rev. B* **81**, 180508(R) (2010).
- ²⁵G. F. Chen, Z. G. Chen, J. Dong, W. Z. Hu, G. Li, X. D. Zhang, P. Zheng, J. L. Luo, and N. L. Wang, *Phys. Rev. B* **79**, 140509(R) (2009).
- ²⁶Z. P. Yin, K. Haule, and G. Kotliar, *Nat. Mater.* **10**, 932 (2011), and supplementary information.
- ²⁷M. Aichhorn, S. Biermann, T. Miyake, A. Georges, and M. Imada, *Phys. Rev. B* **82**, 064504 (2010).
- ²⁸N. L. Wang *et al.*, *J. Phys.: Condens. Matter* **24**, 294202 (2012).
- ²⁹A. A. Schafgans, S. J. Moon, B. C. Pursley, A. D. LaForge, M. M. Qazilbash, A. S. Sefat, D. Mandrus, K. Haule, G. Kotliar, and D. N. Basov, *Phys. Rev. Lett.* **108**, 147002 (2012).
- ³⁰H. Gretarsson *et al.*, *Phys. Rev. B* **84**, 100509(R) (2011).